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1	Electrodeposition of MoS ₂ for Charge Storage in
2	Electrochemical Supercapacitors
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10	
11	Abstract
12	Mo sulfide thin films were cathodically electrodeposited onto glassy carbon electrodes
13	(GCE) from aqueous electrolytes containing 10 mM (NH ₄) ₂ MoS ₄ and 0.2 M KCl. Film adhesion
14	was adequate only for electrodes pretreated by potential cycling in 1.0 M HNO ₃ and 0.1 M NaF to

15 enhance the surface roughness and partially oxidize the GCE. Previous studies report direct cathodic

16 electrodeposition of MoS₂, but energy dispersive x-ray spectroscopy and x-ray diffraction

17 suggest that the as-deposited film is closer in stoichiometry to MoS_3 , which can be converted to

18 MoS_2 by annealing in Ar at 600°C for one hour. The charge storage capability of

19 electrodeposited Mo sulfide films is studied here for the first time in 1.0 M Na₂SO₄ over the

20 thickness range 50 nm to 5 μ m, and before and after high temperature annealing. The highest

21	capacitance is obtained for 50 nm thick MoS_2 films is 330 F/g measured by galvanostatic charge		
22	discharge at 0.75 A/g, and 360 F/g measured by cyclic voltammetry at 10 mV/sec. The		
23	capacitance per unit mass decreases with increasing film thickness due to reduced		
24	electrochemical accessibility. MoS_2 film formed by high temperature annealing in Ar have a		
25	charge storage capability about 40x higher than the as-deposited Mo sulfide films.		
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31	Molybdenum disulfide (MoS_2) is a layered transition metal sulfide that has been extensively		
32	studied for applications such as catalysis, ¹ solid phase lubrication, ² two-dimensional transistors, ³		
33	electronic/spintronic devices, ⁴ materials for intercalation chemistry, ⁵ and electrodes for lithium ion		
34	batteries. ⁶ The unique properties of MoS_2 arise from its two dimensional structure that is analogous to		
35	that of graphene, with strong Mo-S covalent bonds formed within each layer, and weaker van der Waals		
36	bonds between adjacent layers. ^{7,8} MoS_2 thin films have been fabricated predominantly by chemical variables.		
37	deposition (CVD), ^{9,10} mechanical and chemical exfoliation, ^{11,12} hydrothermal synthesis, ^{13,14} microwave		
38	heating, ¹⁵ and solution phase synthesis. ^{16,17} Cathodic MoS_2 electrodeposition has also been reported both		
39	directly from MoS_4^{2-} precursors and from MoO_4^{2-} electrolytes that also contain either thiosulfate or sulfic		
40	as the sulfur source. ¹⁸⁻²² Electrodeposition of MoS_2 is typically simpler and more cost-effective than		
41	CVD and related methods, which involve complex and expensive vacuum technologies. In addition,		
42	electrodeposition yields nm to μ m thick films much more rapidly than exfoliation-based methods.		
43	Compared to other solution phase methods, electrodeposition provides excellent control of film thickness.		
44	Electrodeposition is widely studied for inexpensive scale-up of thin film applications such as photovoltaic		
45	devices and electrochemical supercapacitors. ²³ On the other hand, electrodeposition at low temperature		
46	does not typically yield crystalline or polycrystalline deposits without post-processing.		
47	In electrochemical supercapacitors, thin film metal oxides are often deposited atop high surface		
48	area porous carbon electrodes, ²⁴ which may contain activated carbon, carbon fiber-cloth, carbide-derived		
49	carbon, carbon aerogel, graphite, graphene, or carbon nanotubes. ²⁵ The metal oxide coating stores charge		
50	during electrode polarization by valence change of the metal ion, thus contributing additional		
51	pseudocapacitance to the electrostatic double layer capacitance at the electrode-electrolyte interface. ²⁴		
52	For such applications, desirable properties include chemical stability and corrosion resistance, high		
53	electrical conductivity, widespread abundance, non-toxicity, and high surface area per unit volume and		
54	mass. The most intensively studied metal oxide coatings for supercapacitor applications include RuO ₂ ,		
55	MnO ₂ , and TiO ₂ . ²⁶⁻³⁰		

To date, metal sulfides have attracted less attention than metal oxides as pseudo-capacitive 56 electrode materials within electrochemical supercapacitors. However, chalcogenides such as MoS₂ are 57 intriguing for energy storage applications due to their layered structure, which allows for easier transport 58 and insertion of Li⁺ within battery electrodes, as well as ion transport during oxidation and reduction of 59 metal oxide thin films.^{7,8} However, while MoS₂ within electrochemical supercapacitors has recently been 60 widely studied, these studies have focused not on MoS₂ thin films, but on complex, composite structures 61 62 of nm to µm dimensions in order to optimize the total interfacial active area, and therefore maximize the capacitance per unit mass.^{7,8} Here we report cathodic electrodeposition of MoS₂ atop glassy carbon 63 electrodes, as well as detailed testing of MoS₂ thin film capacitance, both as a function of film thickness 64 65 and before and after high temperature annealing.

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Experimental

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Ammonium tetrathiomolybdate [(NH₄)₂MoS₄], KCl, and NaF were obtained from Acros
Organics. Anhydrous, ACS-grade Na₂SO₄ was obtained from Sigma Aldrich, while concentrated HNO₃
was purchased from Alfa Aesar. Glassy carbon electrodes (GCE) embedded within a Teflon sleeve were
purchased from CH Instruments, and glassy carbon plate was purchased from Alfa Aesar. All reagents
were used as received.

Prior to MoS₂ electrodeposition, the GCE was hand polished with 1.0, 0.3, and 0.05 μm alumina powders. For most experiments, the GCE was subsequently given an electrochemical pretreatment of potential cycling at room temperature between +1.5 and -0.4 V vs. Ag/AgCl at 50 mV/s for 300 cycles in an electrolyte containing 1 M HNO₃ and 0.1 M NaF, and then rinsed with ultrapure water. This procedure is a slight modification of a previously reported method for corrosion studies of glassy carbon electrodes.³¹ This electrochemical pretreatment process was employed here to enhance the surface roughness and partially oxidize the GCE. This pre-treatment process significantly improves film-substrate adhesion, since electrodeposited Mo sulfide films delaminated during capacitance measurements withoutthis pre-treatment.

Glassy carbon electrode (GCE) and glassy carbon plate, with and without MoS₂ coating, were 83 used as the working electrodes in a virgin Teflon three-electrode electrochemical cell with a Pt counter 84 electrode and an Ag/AgCl reference electrode. The GCE had an active area of 0.092 cm², while the 85 glassy carbon plate had an active area of 0.589 cm²prior to annealing and 0.118 cm² after annealing. 86 87 Since the electrochemical cell had to be dissembled following Mo sulfide electrodeposition, and subsequently reassembled following thin film annealing (described below), a smaller O-ring was 88 employed upon cell reassembly to ensure that only substrate regions with an electrodeposited thin film are 89 90 exposed to the electrolyte during subsequent studies. Cyclic voltammetry (CV), galvanostatic charge discharge (GCD), and electrochemical impedance spectroscopy (EIS) measurements were performed with 91 92 a Gamry Instruments Reference 600. EIS measurements over the frequency range 0.01 Hz to 15 kHz 93 employ an AC probe amplitude of 5 mV, and each scan takes about 2.8 min. to acquire.

For some experiments, MoS₂ electrodeposits onto glassy carbon plate were subsequently
annealed in Ar at 600°C for 1 h in a Lindberg Blue M tube furnace. These deposits were then
analyzed by scanning electron microscopy (SEM) and energy dispersive x-ray spectroscopy (EDX) in a
Quanta 450 FEG from FEI Corp, atomic force microscopy in a Nanosurf FlexAFM, and by powder x-ray
diffraction (XRD) in a Rigaku MiniFlex II x-ray diffractometer. Film thickness measurements were
performed by stylus profilometery.

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Results and Discussion

Electrodeposition of Mo sulfide: Figure 1 shows cyclic voltammetry (CV) results for a glassy
 carbon electrode (GCE) in an electrolyte containing 10 mM (NH₄)₂MoS₄ and 0.2 M KCl at pH 6.8.
 By comparison to a blank CV scan, the cathodic peak at approximately -1.0 V vs. Ag/AgCl

corresponds to electrodeposition of a Mo sulfide species. While the research group of Levy-106 Clement first reported cathodic electrodeposition of MoS₂ from MoS₄²⁻,¹⁸⁻²⁰, they did not report 107 cathodic voltammetry peaks such as that observed in Figure 1. However, cathodic peaks varying 108 from about -400 to -600 mV vs. Ag/AgCl were reported for Mo sulfide electrodeposition during 109 successive voltage sweeps through both cathodic and anodic peaks for aqueous solutions 110 containing MoS₄²⁻ without a supporting electrolyte.³² In addition, a cathodic peak was reported at 111 about -1.1 V vs. Ag/AgCl in weakly acidic electrolytes without attribution to a specific cathodic 112 reaction.³³ Electrochemical reduction of Mo sulfide from MoS_4^{2-} -containing electrolytes has 113 been attributed to the following reaction:¹⁸⁻²⁰ 114

115

116
$$MoS_4^{2-} + 4H^+ + 2e^- \to MoS_2 + 2H_2S$$
 (1)

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However, the results from subsequent annealing and EDX analysis of the MoS₂ thin films, combined with powder x-ray diffraction (XRD), suggest that the species formed at the cathodic peak in Figure 1 is closer in stoichiometry to MoS₃ than to MoS₂. On the other hand, since the precursor contains Mo(VI), the cathodic reaction evident in Figure 1 probably cannot directly deposit MoS₃, which also contains Mo(VI). It is possible that reaction (1) is correct, but subsequent chemical/electrochemical reactions result in additional sulfur incorporation into the deposit.

Modest H₂ evolution is visually observed during cathodic electrodeposition of Mo sulfide.
Comparison of the film thickness measured by profilometery and the total charge transferred during
potentiostatic deposition suggests the current efficiency for electrodeposition of Mo sulfide is about 62%.
Potentiostatic electrodeposition at -1.0 V vs. Ag/AgCl allows growth of Mo sulfide films of varying
thickness. Mo sulfide films more than several µm thick contain nm scale cracks. Some Mo
sulfide films were grown on glassy carbon plate and subsequently annealed in Ar at 600°C for 1 h.
After this annealing process, cracks were no longer visible in thicker MoS₂ films.

132 Thin Film Characterization: Following the annealing treatment described above, Figure 2 133 illustrates top-view and cross-sectional SEM images of a MoS_2 thin film that is approximately 5 μ m thick 134 atop a glassy carbon electrode. While such measurements are not fully quantitative, EDX results before and after annealing are consistent with recrystallization of MoS₂ during this process. Prior to annealing, 135 the elemental composition varied, but the elemental Mo:S ratio was always close to 1:3. For annealed 136 films, EDX analyses consistently yield an elemental ratio of Mo:S equal to 1:2 to within experimental 137 138 error. Loss of sulfur during high temperature annealing is consistent with the low boiling point (444.6°C) of elemental sulfur. AFM measurements over a $5x5 \text{ }\mu\text{m}^2$ scan range show that the high temperature 139 annealing process also reduces the rms surface roughness from 231 to 7 nm for a 5 µm thick film. 140 Powder x-ray diffraction (XRD) yields no peaks for the Mo sulfide film prior to high temperature 141 142 annealing. On the other hand, annealed Mo sulfide films yield the XRD pattern shown in Figure 3. This exhibits a broad strong (002) peak at approximately 14.5°, consistent with the formation of crystalline 143 MoS₂.^{34,35} From the (002) peak width in Figure 3, the grain size is estimated as 4.3 nm from the Scherrer 144 equation. However, higher peaks that could distinguish between the 2H and 3R polytypes are not 145 146 observed. Taken together, the EDX and XRD results suggest that the cathodic peak at -1.0 V does not correspond directly to the reaction given in Equation 1, since stoichiometric MoS₂ is only formed after 147 high temperature annealing and loss of a significant amount of sulfur. 148

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150 **Capacitance Testing:** As noted above, the research group of Levy-Clement previously reported cathodic electrodeposition of MoS_2 from $MoS_4^{2-, 18-20}$ but to the best of our knowledge, nobody has studied 151 the charge storage capability of electrodeposited MoS₂ thin films. The charge storage capability of Mo 152 sulfide thin films was assessed, before and after high temperature annealing, as a function of film 153 154 thickness by cyclic voltammetry and galvanostatic charge discharge measurements. During 155 measurements on glassy carbon electrodes (GCE) prepared by mechanical polishing only, the Mo sulfide films delaminated quickly, resulting in a low capacitance value of about 500 μ F/cm² after 100 scans. 156 Delamination occurs due to the poor adhesion between the Mo sulfide thin films and the GCE. Therefore 157

subsequent electrodeposition experiments were preceded by potential cycling between +1.5 and -0.4 V vs. 158 Ag/AgCl for 300 cycles at room temperature in an electrolyte containing 1 M HNO₃ and 0.1 M NaF in 159 order to electrochemically roughen and partially oxidize the surface of the GCE, as described above. 160 To demonstrate the potential applications of electrodeposited Mo sulfide thin films as 161 supercapacitor electrodes, its electrochemical performance was investigated by cyclic 162 voltammetry (CV) in 1.0 M Na₂SO₄ at pH 5.1. Figure 4 illustrates the CV curves for 1.0 µm 163 thick as-deposited MoS₃ and three thicknesses of annealed MoS₂ films at different scan rates 164 ranging from 5 to 100 mV/s over the potential range -200 to +400 mV vs. Ag/AgCl reference 165 electrode. The results in Figure 4 exhibit no cathodic or anodic current peaks, so the Mo sulfide 166 films studied here are close to ideal pseudocapacitive materials, albeit over a modest potential 167 window of ~600 mV.³⁶ From these cyclic voltammograms, the capacitance (C) per unit mass 168 169 (m) can be determined according to:

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171
$$C = \frac{S}{2mk(U_2 - U_1)}$$
 (2)

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173 where S is the area enclosed by each curve, k is the scan rate, and $U_2 - U_1$ is the potential range scanned. 174 The capacitance per unit area (A) can also be obtained from Equation (2) by replacing mass with area. Although studies of thin film metal capacitors often report the capacitance per unit mass, capacitance per 175 176 unit area is a more fundamental measure of charge storage capability. Such comparisons allow 177 quantitative determination of the value of extra MoS₂ film thickness for increasing the electrode capacitance. One might expect thin film pseudo-capacitance to be limited to only a thin film region near 178 179 the electrode surface due to transport limitations for ion penetration into the bulk electrode material. Comparing Figures 4a and 4c, the specific capacitance of the as-deposited MoS₃ is about 40x 180 181 lower than that of the annealed MoS₂ films, so further analysis was performed only on the latter. The low capacitance of the as-deposited film is consistent with deposition of an amorphous film of non-182

stoichiometric Mo sulfide. The improved film pseudocapacitance after annealing can be attributed to the
layered chalcogenide structure, where the relatively weak interlayer bonding provides mechanical
flexibility to accommodate ion transport during electrochemical reactions.^{7,8}

Figure 5 summarizes the results for capacitance testing over 1000 cycles of MoS₂ films from 50 186 nm to 5 μ m thick in 1.0 M Na₂SO₄ at a scan rate of 10 mV/sec. As the film thickness increases from 50 187 nm to 5 µm, the capacitance per unit mass in Figure 5 decreases continuously. This is consistent with 188 previous studies of transition metal oxide pseudocapacitance that suggest a penetration depth of 20-50 nm 189 for protons and other species involved in charge storage.³⁷⁻³⁹ Material further below the solid-electrolyte 190 interface becomes electrochemically inaccessible, and this is thus at least partly inactive for charge 191 storage. This suggests that the best capacitance obtained here (350-400 F/g at 10 mV/sec) for 50 nm thick 192 MoS₂ cannot be increased much further by electrodeposition of thinner films. Figure 5 illustrates that 193 after 1000 cycles, 50 nm MoS₂ films retain 87% of their original capacitance, and films 250 nm and 194 thicker films retain 90-100% of their original capacitance. Figure 6 illustrates the effect of scan rate on 195 the measured capacitance for a range of MoS₂ film thickness. The decrease in film capacitance with 196 increasing scan rate is often observed for MoS₂ and other materials tested within electrochemical 197 supercapacitors,⁴⁰⁻⁴² and is expected for the current study due to the relatively thick (µm range) films 198 199 tested. However MoS₂ electrodes thicker than 50 nm demonstrate excellent capacitance performance at high scan rate. For instance, the specific capacitance, as shown in Figure 6, of 125 nm thick MoS₂ 200 201 electrode at scan rate of 100 mV/s is about 80% comparing with that at 10 mV/s, and this percentage 202 increases with film thickness.

MoS₂ film capacitance was also determined by galvanostatic charge discharge (GCD)
measurements, with Figure 7 illustrating the GCD results for three different MoS₂ film thickness: 50 nm,
1 μm, and 5 μm. The GCD curves are close to the triangular shape expected for ideal pseudocapacitive
behavior, except for the lowest current densities studied.^{43,44} The capacitance can be precisely
determined from the GCD curves using the equation:⁴⁵

209
$$C = \frac{i}{m\frac{dU}{dt}}$$



212 Figure 8 illustrates the effect of charge/discharge current density on the MoS₂ film capacitance 213 calculated from GCD measurements. The decline in film capacitance with increasing current density in 214 Figure 8 reflects the same underlying phenomena as the decline in capacitance with increasing scan rate 215 in Figure 6, reduced energy storage capability for high power applications. Similar to what is observed 216 for specific capacitance vs. scan rate in Figure 6, MoS₂ electrodes thicker than 50 nm demonstrate good 217 capacitance performance at high current density. Specifically, the specific capacitance of 125 nm thick MoS₂ electrode at current density of 5 A/g is about 80% comparing with that at 1 A/g. Figure 9 218 219 summarizes the capacitance measurements obtained for MoS_2 films of different thickness by 220 galvanostatic charge discharge. Figure 9 illustrates that capacitance retention ranges from 74-91% for the different MoS₂ film thickness studied. The current densities at which each curve is given in Figure 9 are 221 222 chosen for a close correspondence to Figure 5a. 223 The highest capacitance per unit mass obtained for annealed MoS_2 is observed for 50 nm thick films, and ranges from 350-400 F/g from both cyclic voltammetry at 10 mV/sec (Figure 5a) and 224 galvanostatic charge discharge at 0.75 A/g (Figure 9). These values can be compared to other reports of 225 226 MoS_2 incorporation into electrochemical supercapacitors, but often the materials tested do not correspond 227 to bulk MoS₂ films such as those electrodeposited here, so these comparison must be carefully made. For 228 example, many capacitance measurements on pure MoS_2 involves materials with only 1-3 monolayers 229 thick, whose electronic properties differ substantially from those of bulk MoS₂. In addition, many researchers have studied composite materials that are not purely MoS₂, and their results depend on 230 231 composition, structure, grain size, and porosity, as well as scan or charging rate. 232 The authors are aware of only three studies of the capacitance of relatively thick, purely MoS_2

thin films.⁴⁰⁻⁴² Hydrothermal synthesis of porous MoS_2 thin films has been reported with capacitance up

to 403 F/g.³⁹ Magnetron sputtering has been employed to grow porous MoS₂ films of capacitance 330 234 F/cm^{3} ⁴¹ In addition, MoS₂ and graphene nanofilms were grown separately by exfoliation, and then 235 combined into thin film geometry by co-precipitation, yielding film capacitance values as high as 13 236 mF/cm².⁴² In summary, the best capacitance values obtained here (350-400 F/g) for 50 nm MoS₂ films 237 are similar to the results from references #40-42 for relatively thick MoS₂ films. Higher capacitance 238 239 values ranging from 416-589 F/g have been reported for MoS₂ nanocomposites with reduced graphene oxide,⁴³ carbon nanotubes,⁴⁶ porous carbon,⁴⁷ polyaniline,⁴⁸ and WS₂ and amorphous carbon.⁴⁹ In 240 addition, much higher capacitance values up to 1544 F/g have been reported for MoS₂-Ni₂S₃ 241 nanocomposites atop porous C and Ni electrodes.50,51 242

The nature and stability of annealed MoS₂ thin films of different thickness were further 243 investigated by electrochemical impedance spectroscopy (EIS) in 1.0 M Na₂SO₄. EIS studies are 244 sometimes employed to verify that materials intended for charge storage within electrochemical 245 supercapacitors are truly pseudocapacitive, and thus can be charged and discharged rapidly.³⁶ The results 246 for a 1.0 µm thick annealed MoS₂ film, before and after capacitance testing by cyclic voltammetry, are 247 shown in Figure 10. In these Nyquist plots, the real component of the impedance is plotted on the x-axis 248 249 and the imaginary component on the y-axis. For materials that undergo redox reactions, the Nyquist plot exhibits a semicircular shape at intermediate to high frequencies, indicating relatively slow charge 250 transfer. The results in Figure 10 are consistent with truly pseudocapacitive materials, with the expected 251 capacitive behavior approaching a vertical line.⁴³ The results for other MoS₂ film thickness are almost 252 253 identical.

In addition, the results of Figure 10 suggest that the electrical properties of the MoS_2 thin film are largely unaffected by capacitance testing. However, the slight reduction in the x-intercept likely indicates that the MoS_2 film resistance decreases slightly during capacitance testing. Although the difference in the x-intercepts is small, this is a real affect, since the electrochemical cell was not disturbed during these measurements, so the electrode positions remain fixed.

Conclusions

262	Mo sulfide thin films can be cathodically electrodeposited onto glassy carbon from aqueous
263	electrolytes containing 10 mM (NH ₄) ₂ MoS ₄ and 0.2 M KCl at pH 6.8. EDX measurements yield a Mo:S
264	elemental ratio of about 1:3 in the as-deposited films, but this changes to 1:2 after annealing at 600°C in
265	Ar for one h. Similarly, the as-deposited Mo sulfide films do not exhibit any XRD peaks, but a broad
266	MoS ₂ (002) peak is observed after high temperature annealing. From the Scherrer equation, the grain size
267	is estimated as 4.3 nm. For the first time, the charge storage capability of electrodeposited Mo sulfide
268	films is studied by potential scanning and galvanostatic charge discharge measurements in 1.0 M Na ₂ SO ₄
269	over the thickness range 50 nm to 5 μm . The highest capacitance obtained ranges from 350-400 F/g for
270	50 nm thick MoS_2 films. MoS_2 film formed by high temperature annealing in Ar have a charge storage
271	capability about 40x higher than the as-deposited Mo sulfide films.

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355		Figure captions
356	Figure 1.	Cyclic voltammetry of a glassy carbon electrode (GCE) at 30 mV/sec with and without
357		10 mM (NH ₄) ₂ MoS ₄ in 0.2 M KCl at pH 6.8.
358	Figure 2.	Top view (a) and cross-sectional (b) SEM images of 5 μ m thick MoS ₂ thin films atop
359		glassy carbon plate after annealing at 600°C in Ar for 1 h.
360	Figure 3.	Powder x-ray diffraction of MoS_2 thin films scraped off glassy carbon plate after
361		annealing at 600°C in Ar for 1 h.
362	Figure 4.	Cyclic voltammetry curves at different scan rates in $1.0 \text{ M} \text{ Na}_2 \text{SO}_4$ for $1 \mu\text{m}$ as-deposited
363		film (a); and 50 nm (b), 1 μm (c) and 5 μm (d) annealed MoS_2 thin film atop glassy
364		carbon plate.
365	Figure 5.	Capacitance measurements from cyclic voltammetry at 10 mV/s scan rate for different
366		thickness of annealed MoS_2 thin films in units of F/g (a) and F/cm ² (b) during 1000
367		cycles in 1.0 M Na ₂ SO ₄ .
368	Figure 6.	Specific capacitance measured for first cycle as a function of scan rate and film thickness
369		for annealed MoS_2 films.
370	Figure 7.	Galvanostatic charge/discharge curves of 50 nm (a), 1μ m (b) and 5 μ m (c) annealed
371		MoS ₂ films at different charge/discharge current densities.
372	Figure 8.	Effect of charge/discharge current density on the specific capacitance obtained for
373		different MoS ₂ film thickness.
374	Figure 9.	Capacitance measurements from galvanostatic charge/discharge technique for different
375		thickness of MoS_2 thin films during 1000 cycles in 1.0 M Na_2SO_4 .
376	Figure 10.	Nyquist plot of EIS results in 1.0 M Na ₂ SO ₄ for MoS ₂ film atop glassy carbon plate,
377		before and after capacitance testing.



_ _ _



- 384 Figure 1.





Figure 2a.







417

419 Figure 4a.

-0.2

-0.1

0

0.1

Potential (V vs. Ag/AgCl)

0.2

0.3

0.4

420

421

422



428 Figure 4b.





442 Figure 4c.





Potential (V vs Ag/AgCl)

Figure 4d.











483 Figure 6.















4.5





536 Figure 9.



